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Zhihui Yinª; Yajie Zhang^ь; Zhang Xiaominª; Jinghua Yinª

a Polymer Physics Laboratory, Changchun Institute of Applied Chemistry of Chinese Academy Sciences, Changchun, P.R. China ^b Department of Applied Chemistry, Jilin University of Technology, Changchun, P.R. China

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Preparation and Characterization of Melt Grafted Acrylamide Tertiary Butyl Sulfonic Acid onto Polypropylene

ZHlHUl YIN a, YAJIE ZHANG ', ZHANG XIAOMIN **^a** and JINGHUA YIN^a

^a Polymer Physics Laboratory, Changchun Institute of Applied Chemistry *of Chinese Academy Sciences, Changchun 130022 P.R. China; b*Department of Applied Chemistry, Jilin University of Technology, *Changchun 130022 P. R. China*

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Grafting of acrylamide tertiary butyl sulfonic acid (ATBS) onto polypropylene (precisely, ethylene/propylene random copolymer containing 8% ethylene content) was carried out by melt extruding with addition of dicumyl peroxide (DCP) as **an** initiator. The structure of graft copolymer was characterized by using WAXS, **FT-IR, ESCA** and **DSC.** Improved thermal stability was observed for *graft* copolymer. **The** effect of monomer and initiator concentration, reactive temperature and srew speed on degree of grafting efficiency of grafting has been investigated.

Keywords: Acrylamide; butyl sulfonic acid; polypropylene melt grafting

INTRODUCTION

Grafting copolymers have been widely used for the compatibilization of immiscible polymer blend as well as the improvement of interfacial adhesion in polymeric composites $[1 \sim 2]$. In comparison with traditional copolymerization processes, the modification of existing polymer **by** reactive melt processing offers not only the flexibility in tailoring polymer properties for specific application but also reduction in equipment **[3].** Consequently, reactive polymer processing has received increasing attention polymer scientist and engineers $[4 \sim 10]$. For example, the grafting reaction of maleic anhydride onto molten polyethylene [7] and polypropylene [8] (PP) has been previously studied. However. the graft reaction of acrylamide tertiary butyl sulfonic acid (ATBS) onto PP by means of a melt process has not found to be reported. The sulfonic acid often interact strongly with other functional groups. which leads to miscibility enhancement.

In the present paper, the melt grafting reaction of ATBS onto PP has been investigated. The purpose is to use the functional PP as the effective toughening agent of polyamide and a **in-situ** compatibilizer of polyamide-polypropylene blend. The degree of grafting was determined by means of element analysis. FT-lR, WAXS, DSC and ECSA were also applied to characterize structure of graft copolymer.

EXPERIMENTAL

Materials

The polymers used in this work were a ethylene/propylene random copolymer containing 8% ethylene (PP1330) manufactured by YanShan Petrochemicals Corp. LTD. (China), and with melt index 1.5 g/10 min. Acrylamide tertiary butyl sulfonic acid (ATBS) and dicumyl peroxide (DCP) were reagent grade.

Preparation of PP-g-ATBS

A Brabender single screw extruder was used to prepare PP-g-ATBS. Composition of reaction mixture was 500g of PP, 0 to 5 wt% of ATBS and 0 to 1.0 wt% of DCP. The screw speed was $30 \sim 80$ rpm. Temperature of extruding process was $185 \sim 225$ °C. The graft copolymers were compression molded into thin films, and then extracted with alcohol for 48 hours to remove traces ATBS and homopolymer. Finally, films were dried under reduced pressure at 80° C to constant weight.

Characterization of PP-g-ATBS

The IR spectra of purified films were recorded onto a FTS-7 Fouriertransform infrared spectrometer.

The ESCA spectra of purified films were recorded onto an ESCALAB-MK2 electron energy spectrometer.

Differential scanning calorimetric studies were conducted using a Perkin-Elmer 7 thermal analyzer. The analysis was carried out at a constant heating rate of 10° C/min in the temperature range of 25-200°C under nitrogen atmosphere.

The WAXS spectra of purified films were performed on a HP1700 wide angle X-ray diffraction spectrometer with Cu K α , Ni-filtered radiation.

The content of ATBS in graft copolymer was determined by using element analysis instrument.

Contact angle(θ) of a water drop on PP and PP-g-ATBS samples were measured at 20° C by using a home-made apparatus.

Melt flow index were determined using an extrusion plastometer (JiLin University Scientific Instruments Inc., China) standardized according to ASTM D1238-79. The temperature was set at 230° C, and 2.16 kg load.

RESULTS AND DISCUSSION

Characterization of Graft Copolymer

FTIR spectroscopy is an well known method to identify functional groups and to make quantitative analysis. The FTIR spectra of PP and PP-g-ATBS are shown in Figure 1. The appearance of two peaks in the purified graft sample at 1718 and 1665 cm^{-1} indicates the addition of amido groups in PP molecular backbone. It was shown that ATBS monomer has been grafted onto PP molecular chains. Finer differences were observed in the 721 cm^{-1} band which is attributed to the rocking-mode vibration of long ethylene sequences- $(CH₂)_n$ -with *n* [11]. A lowering of the absorbance with increasing the degree of grafting. This feature suggests that ATBS monomer may be mainly grafted onto the ethylene sequences of the polymer molecular chains. Figure2 is the **ESCA** spectra of PP and purified graft copolymer. In the graft sample, the O_{1s} , N_{1s} and S_{2n} peaks appeared at 532ev, 399.9ev and 167.8ev, respectively. The PP spectra has only C_{1s} peak at 284.6ev. This result verified that ATBS has been grafted onto PP molecular chains.

FIGURE 1 The FTIR spectra of PP (a) PP-g-ATBS(b,c,d); (b)-containing ATBS 0.26%: (c)-containing ATBS 0.42% (d)-containing ATBS 0.78%.

FIGURE 2 The ESCA spectra of PP(a) and PP-g-ATRBS and (b).

The **DSC** cooling curves of PP and graft copolymer are given in Figure **3.** Plain PP sample shows abroad exotherm, its crystallization temperature of ethylene sequences is closed to that of propylene sequences, the two crystallization peak overlapped. With the increase of the degree of grafting, the crystallization exothermic peak of ethylene sequences decreased gradually. This indicated that the grafted **ATBS** damaged the regularity of ethylene sequences. The crystallization temperature of propylene sequences increased with increasing degree of grafting. The observed increase in crystallization temperature can be

FIGURE 3 The DSC cooling curves of PPm (a) and PP-g-ATBS(b, c, d); (b) containing ATBS 0.19%; (c) containing ATBS 0.47%; (d) containing ATBS 0.88%.

attributed to that ATBS act as a nucleating agent in propylene sequences crystallization process. The WAXS spectra of PP and PP-g-ATBS are shown in Figure 4. As indicated by the patterns, grafting ATBS leads to some change of crystallization structure of PP crystals. The peak at $2\theta = 15.9$ degree, which corresponds to the 300 diffraction plane of the β form (hexagonal), decreased with increase of the degree of grafting of PP-g-ATBS. The relative amount of the β -form is usually described in terms of the K value, which is defined as:

 $K = \frac{H(300)}{H(300) + H(110) + H(040) + H(130)}$

where $H(110)$, $H(040)$ and $H(130)$ are the heights of three strong equatorial α -form peaks (110), (040) and (130), and H(300) the height of the strong single β -form peak (300). Using this equation, the **K** value of PP, pp-g-ATBS of ATBS content 0.26%,0.57% and 0.78% as follows: 28.3%, 17.4%, 12.1% and 0.57%, respectively. It may be concluded that, by grafting of ATBS on PP, the proportions of monoclinic $(\alpha$ -form) and hexagonal $(\beta$ -form) phases is modified in the PP. The proportion

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FIGURE 4 The WAXD spectra of PP (a) PP-g-ATBS: (b, c, d); (b) containing ATBS 0.26%; (c) containing ATBS 0.57%; (d) containing ATBS 0.78%.

of these tno forms usually depends on preparing conditions of samples and nucleating process. In our work preparing conditions were same for all samples. Therefore. it can be concluded that the change of the crystallization structure of PP-g-ATBS was originated from a different nucleating process.

The content angle (θ) of graft copolymer films were measured for different solvents of variable surface tensions. From the results obtained, the critical surface tension (y_c) values of grafted samples were calculated by extrapolating the surface tension against $\cos\theta$ plot to $\cos\theta = 1$. The critical surface tension(γ_c) of the PP and PP-g-ATBS samples is listed in Table **I.** The result suggested that the surface polarity of graft copolymer is augmented with an increase of the content of ATBS in PP-g-ATBS samples.

Influence of Monomer (ATBS) and Initiator (DCP) Concentration

Table I shows the influence of monomer and initiator concentration on the grafting reaction at a reaction temperature of 200° C. Run 1-5

Run	monomer concentration $(wt\%)$	initiator concentration (wt%)	degree of grafting $(wt\%)$	MFI MFI (a/10min)	γ_c γ. (dyn/cm)
		0.05	0.26	1.8	29.8
		0.1	0.34	2.4	27.3
		0.5	0.78	6.0	24.5
4		1.0	0.89	114	22.7
		0.25	0.57	3.8	25.9
6	3.5	0.25	0.44	3.6	26.1
	2.5	0.25	0.31	3.8	30.1
9	0.5	0.25	0.19	3.8	31.2

TABLE I Influence of monomer and initiator concentration on graft reaction

All runs carried out at **200°C** for **60rpm.**

showed the influence of initiator concentration for a fixed monomer concentration of *5* wt%. As the initiator concentration was increased from 0.05 to 1 wt%, the degree of grafting increased sharply. The melt flow index (MFI), which is an indication of the extent of degradation, increased with increasing initiator concentration, indicated a increase in degradation of PP.

Run 5-9 showed the influence of monomer concentration on the grafting reaction for a fixed initiator concentration of 0.25 wt\% . As the monomer concentration was decreased from 5 wt% to 0.5 wt%, the degree of grafting decreased marginally, i.e. from 0.57 to 0.19 wt%. The MFI values were unaffected by changing the monomer concentration.

Influence of Screw Speed and Reactive Temperature

Table **I1** exhibits the effect of screw speed on the degree of grafting. It was observed that the degree of grafting increases with speed decreased and then remains constant. With an increase in reaction time, the radicals will have more time for'reaction, and as a result, the degree of grafting will increase. After some time all initiator was used up, and no further change in degree of grafting was observed with an increase in reaction time. The effect of reaction temperature on degree of grafting was listed in Table **11.** With increasing reaction temperature, the degree of grafting increases from 185° C to 200° C, and the degree of grafting reached a maixmum value at 200°C. However beyond 200"C, it decreases with increase in temperature. With increasing temperature,

Run	temperature (°C)	screw speed (rpm)	degree of grafting $(wt\%)$	MFI (q/10min)
10	185	60	0.23	3.2
11	195	60	0.42	3.6
12	200	60	0.57	3.8
13	210	60	0.69	3.8
14	225	60	0.47	4.1
15	200	30	0.88	3.8
16	200	45	0.84	3.9
17	200	80	0.79	4.0

TABLE11 graft reaction Influence of screw speed and temperature on

All runs **are** *5* **wt% ATBS** and *0.25* **wt% DCP.**

degree of grafting increases due to increase decomposition rate of initiator. Therefore, the number of free radicals and their mobility also increase resulting in a **high** degree of grating high mobility also results in mutual termination of free radicals for copolymerization. Hence, the maximum value of degree of grafting was appeared. From Table **11,** it was seen that the reaction time and temperature has only a small effect on the **MFI.**

CONCLUSIONS

The results show that **ATBS** monomer had been grafted onto PP by melt mechanical mixing. Various reaction condition that affected the degree of grafting were investigated. The initiator concentration largely controlled the degree of grafting and melt flow index, and the screw speed and temperature had a small effect on the MFI. The degree of grafting increased with increasing monomer concentration, but the **MFI** remained constant. Crystallization temperature of grafting copolymer was higher than that of **PP,** and the crystal form of grafting sample was different from the PP.

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